# 2. Crystal Structure

crystalline solid – the atoms or ions arrange in a pattern that repeats itself in three dimensions to form a solid which has long-range order amorphous solid – materials with only short-

range order

space lattice – a network composed of an infinite
three-dimensional array of points

unit cell – the repeating unit in a space lattice

#### space lattice

unit cell





lattice constants lattice vector – a, b, cinteraxial angle –  $\alpha$ ,  $\beta$ ,  $\gamma$ 

# crystal system

# only 7 different types of unit cells 14 standard (Bravais) unit cells could describe all possible lattice networks

crystal	axial lengths &	Nakan Provident
system	interaxial angles	space lattice
cubic	a = b = c	simple cubic
	$\alpha = \beta = \gamma = 90^{\circ}$	body-centered cubic
		face-centered cubic
tetragonal	$a = b \neq c$	simple tetragonal
	$\alpha = \beta = \gamma = 90^{\circ}$	body-centered tetragonal
orthorhombic	$a \neq b \neq c$	simple orthorhombic
	$\alpha = \beta = \gamma = 90^{\circ}$	body-centered orthorhombic
		base-centered orthorhombic
and the second	a weather a	face-centered orthorhombic
rhombohedral	a = b = c	simple rhombohedral
	$\alpha = \beta = \gamma \neq 90^{\circ}$	
hexagonal	$a = b \neq c$	simple hexagonal
	$\alpha = \beta = 90^{\circ}$	
	$\gamma = 120^{\circ}$	
monoclinic	$a \neq b \neq c$	simple monoclinic
、 第119年前十年。	$\alpha = \gamma = 90^{\circ} \neq \beta$	base-centered monoclinic
triclinic	$a \neq b \neq c$	simple triclinic
	$\alpha \neq \beta \neq \gamma \neq 90^{\circ}$	Sector and the sector of the sector



metallic crystal structures 90% elemental metals crystallize into three crystal structures:

• body-centered cubic (BCC)			
Statistics in	a (nm)	R (nm)	
Cr	0.289	0.125	
Fe	0.287	0.124	
Mo	0.315	0.136	
K	0.533	0.231	
Na	0.429	0.186	
Та	0.330	0.143	
W	0.316	0.137	
V	0.304	0.132	

• face-centered cubic (FCC)		
	a (nm)	R (nm)
Al	0.405	0.143
Cu	0.3615	0.128
Au	0.408	0.144
Pb	0.495	0.175
Ni	0.352	0.125
Pt	0.393	0.139
Ag	0.409	0.144







### • hexagonal close-packed (HCP)

a	<i>c</i> (nm)	R (nm)
0.2973	0.5618	0.143
0.2665	0.4947	0.133
0.3209	0.5209	0.160
0.2507	0.4069	0.125
0.3231	0.5148	0.160
0.2950	0.4683	0.147
0.2286	0.3584	0.113
	<i>a</i> 0.2973 0.2665 0.3209 0.2507 0.3231 0.2950 0.2286	ac (nm)0.29730.56180.26650.49470.32090.52090.25070.40690.32310.51480.29500.46830.22860.3584



coordination number = 8 total 2 atoms per unit cell  $\sqrt{3}a = 4 R$   $a = \frac{4 R}{\sqrt{3}}$ atomic packing factor (APF) APF = volume of atoms in unit cell APF = 0.68

FCC



coordination number = 12 total 4 atoms per unit cell  $\sqrt{2}a = 4 R$   $a = \frac{4 R}{\sqrt{2}}$ 

APF = 0.74

the closest packing possible of spherical atoms cubic closest-packed

НСР



coordination number = 12 total 2 atoms per unit cell APF = 0.74

the closest packing possible of spherical atoms

c/a ratio for ideal HCP structure is 1.633



atom positions in cubic unit cell



BCC unit cell coordinates of eight corners: (0, 0, 0) (1, 0, 0) (0, 1, 0) (0, 0, 1)(1, 1, 0) (1, 0, 1) (0, 1, 1) (1, 1, 1)coordinate of the center:  $(\frac{1}{2}, \frac{1}{2}, \frac{1}{2})$ 

direction index – the vector components of the direction resolved along each of the coordinate axes and reduced to the smallest integers



all parallel direction vectors have the same direction indices

directions are crystallographically equivalent if the atom spacing along each direction is the same

ex. cubic edge directions:

 $\begin{bmatrix} 100 \\ 010 \end{bmatrix} \begin{bmatrix} 001 \\ 001 \end{bmatrix} \begin{bmatrix} 010 \\ 001 \end{bmatrix} \begin{bmatrix} 100 \\ 001 \end{bmatrix} \begin{bmatrix} 100 \\ 100 \end{bmatrix}$ 

equivalent directions are called indices of a family or form

ex. draw the following directions:

(a) [112]

(b) [110]



(c)  $[32\bar{1}]$ 



Miller indices for crystallographic planes Miller notation system (*hkl*)

Miller index – the reciprocals of the fractional intercepts that the plane makes with the *x*, *y*, and *z* axes of the three nonparallel edges of the cubic unit cell

procedure for determining Miller index:

- (1) choose a plane not pass through (0, 0, 0)
- (2) determine the intercepts of the plane with *x*, *y*, and *z* axes
- (3) form the reciprocals of these intercepts
- (4) find the smallest set of whole numbers

that are in the same ratio as the intercepts



ex. draw the following crystallographic planes in cubic unit cell:





# planes of a family or form $\{hkl\}$ ex. (100), (010), (001) are a family $\{100\}$

an important relationship for cubic system, the direction indices of a direction perpendicular to a crystal plane are the same as the Miller indices of that plane interplanar spacing between two closest parallel planes with the same Miller indices is designated  $d_{hkl}$  (*h*, *k*, *l* are the Miller indices)



 $d_{hkl} = \frac{a}{\sqrt{h^2 + k^2 + l^2}}$ 

a: = lattice constant

ex. determine the Miller indices of the planes shown as follow:



(5120)



 $(\bar{6}\bar{4}6)$ 

#### hexagonal structure

Miller-Bravais indices – HCP crystal plane indices (*hkil*) h + k + i = 0three basal axes  $a_1, a_2, a_3$  and c axis



basal planes<br/>prism planes<br/>(ABCD)(0001)<br/>(1010)(ABCD)(1010)(ABEF)(1100)(CDGH)(0110)





direction indices in HCP unit cell four indices [uvtw] u + v + t = 0 $u = \frac{1}{3}(2u - v)$   $v = \frac{1}{3}(2u - v)$  t = -(u + v)directions  $a_1, a_2, a_3$  $-a_1$  $+a_2$ 1210]  $+a_3$  $-a_3$ [1120] [2110]  $-a_2$  $+a_1$  $+a_3$  direction incorporating c axis [1121] [1210]  $+a_2$  $+a_3$ 2110] [1120]

#### directions on the upper basal planes



volume, planar, linear density volume density

#### mass/unit cell

# $\rho_v = \frac{1}{\text{volume/unit cell}}$

ex. Cu has FCC structure, atomic radius of 0.1278 nm, atomic mass of 63.54 g/mol calculate the density of Cu in Mg/m<sup>3</sup>.

FCC structure  $\sqrt{2} a = 4 R$   $a = 2 \sqrt{2} R = 2 \sqrt{2} (1.278 \times 10^{-10})$   $= 3.61 \times 10^{-10} \text{m}$   $V = (3.61 \times 10^{-10} \text{ m})^3 = 4.70 \times 10^{-29} \text{ m}^3$ 4 Cu per unit cell  $m = 4 \times 63.54 \times 1.66 \times 10^{-30} \text{ Mg} = 4.22 \times 10^{-28} \text{ Mg}$   $\rho_v = 4.22 \times 10^{-28} \text{ Mg} / 4.70 \times 10^{-29} \text{ m}^3$  $= 8.98 \text{ Mg/m}^3 \text{ (exp.} = 8.96 \text{ Mg/m}^3 \text{)}$ 

planar atomic density

 $\rho_p = -$ 

equiv. no. of atoms whose centers are intersected by selected area

#### selected area

ex. calculate planar atomic density  $\rho_p$  on (110) plane of the  $\alpha$ -Fe in BCC lattice in atoms/mm<sup>2</sup>. (lattice constant a = 0.287 nm)



1 atom (center) + <sup>1</sup>/<sub>4</sub> atom (corner) × 4 = 2 atoms area =  $a \times \sqrt{2} a = \sqrt{2} a^2 = \sqrt{2} (2.87 \times 10^{-7})^2$ = 1.164 × 10<sup>-13</sup> mm<sup>2</sup>  $\rho_p = \frac{2 \text{ atoms}}{1.164 \times 10^{-13} \text{ mm}^2} = 1.72 \times 10^{13} \text{ atoms/mm}^2$ 

linear atomic density

 $\rho_1 =$ 

no. of atoms diam. intersected by selected length of line in direction of interest

selected length of line

ex. calculate linear atomic density  $\rho_l$  in [110] direction in Cu crystal lattice in atoms/mm. (Cu is FCC and lattice constant a = 0.361 nm) no. of atoms =  $\frac{1}{2} + 1 + \frac{1}{2} = 2$  atoms length =  $\sqrt{2} a = \sqrt{2} (3.61 \times 10^{-7})$ = 5.104 × 10<sup>-7</sup> mm 2 atoms  $\rho_l = \frac{2}{5.104 \times 10^{-7}} = 3.92 \times 10^{6}$ 



### polymorphism or allotropy

element or compound exists in more than one crystalline form under different conditions of temperature and pressure

ex.

	crystal structure	at other
metal	at room temperature	temperature
Ca	FCC	BCC (> 447°C)
Co	НСР	FCC (> 427°C)
Hf	НСР	BCC (> 1742°C)
Fe	BCC $(\alpha)$	FCC (912-1394°C) ( <i>y</i> )
		BCC (>1394°C) (δ)
Li	BCC	HCP (< -193°C)
Na	BCC	HCP (< -233°C)
T1	НСР	BCC (> 234°C)
Ti	НСР	BCC (> 883°C)
Y	НСР	BCC (> 1481°C)
Zr	НСР	BCC (> 872°C)

crystal structure analysis

#### X-ray sources

x-rays used for diffraction are radiations with wavelengths  $0.05 \sim 0.25$  nm



a voltage of 35 kV is applied between cathode (W filament) and anode (Mo target) x-ray spectrum  $0.2 \sim 1.4$  nm wavelength of K<sub>a</sub> line 0.07 nm



X-ray diffraction reflected wave patterns of beam are not in phase, no reinforced beam will be produced destructive interference occurs



reflected wave patterns of beam are in phase, reinforcement of the beam or constructive interference occurs



ex. BCC Fe placed in an x-ray diffractometer using x-ray with  $\lambda = 0.1541$  nm. diffraction from {110} planes was obtained at  $2\theta =$ 44.704°. calculate lattice constant *a*.

 $\lambda = 2 d_{110} \sin \theta$  $d_{110} = \frac{\lambda}{2\sin \theta} = \frac{0.1541 \text{ nm}}{2\sin(22.352^\circ)} = 0.2026 \text{ nm}$  $a = d_{hkl} \sqrt{h^2 + k^2 + l^2} = 0.2026\sqrt{2} = 0.287 \text{ nm}$ 

X-ray diffraction analysis of crystal structures powder diffraction method

diffractometer





diffraction pattern for cubic unit cell

 $d_{hkl} = \frac{a}{\sqrt{h^2 + k^2 + l^2}} \text{ and } \lambda = 2d \sin\theta$  $\lambda = \frac{2a \sin\theta}{\sqrt{h^2 + k^2 + l^2}}$ 

rules for determining the diffracting {*hkl*} planes in cubic crystals

reflection presentreflection absentBCC(h + k + l) = even(h + k + l) = oddFCC(h, k, l) all odd or(h, k, l) not all oddall evenor all even

ex. Diffraction pattern for W sample by the use of a diffractometer with Cu radiation



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W : BCC structure

$$\lambda = \frac{2 a \sin \theta}{\sqrt{h^2 + k^2 + l^2}}$$
$$\sin^2 \theta = \frac{\lambda^2 (h^2 + k^2 + l^2)}{4a^2}$$
$$\frac{\sin^2 \theta_{\rm A}}{\sin^2 \theta_{\rm B}} = \frac{h^2_{\rm A} + k^2_{\rm A} + l^2_{\rm A}}{h^2_{\rm B} + k^2_{\rm B} + l^2_{\rm B}}$$

Miller indices of the diffracting planes for BCC and FCC

{hkl}	$\Sigma[h^2 + k^2 + l^2]$	FCC	BCC
{100}	1		
{110}	2		110
{111}	3	111	
{200}	4	200	200
{210}	5		
{211}	6	· · · · · ·	211

first two sets of diffraction planes FCC {111) and {200}  $\frac{\sin^2 \theta_A}{\sin^2 \theta_B} = \frac{h^2_A + k^2_A + l^2_A}{h^2_B + k^2_B + l^2_B} = 0.75$ 

