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Magnetic control of ferroelectric polarization

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The magnetoelectric effect—the induction of magnetization by means of an electric field and induction of polarization by means of a magnetic field—was first presumed to exist by Pierre Curie¹, and subsequently attracted a great deal of interest in the 1960s and 1970s (refs 2–4). More recently, related studies on magnetic ferroelectrics^{5–14} have signalled a revival of interest in this phenomenon. From a technological point of view, the mutual control of electric and magnetic properties is an attractive possibility¹⁵, but the number of candidate materials is limited and the effects are typically too small to be useful in applications. Here we report the discovery of ferroelectricity in a perovskite manganite, TbMnO₃, where the effect of spin frustration causes

sinusoidal antiferromagnetic ordering. The modulated magnetic structure is accompanied by a magnetoelastically induced lattice modulation, and with the emergence of a spontaneous polarization. In the magnetic ferroelectric TbMnO₃, we found gigantic magnetoelectric and magnetocapacitance effects, which can be attributed to switching of the electric polarization induced by magnetic fields. Frustrated spin systems therefore provide a new area to search for magnetoelectric media.

The room-temperature crystal structure of TbMnO₃ investigated here is the orthorhombically distorted perovskite structure (space group *Pbnm*; Fig. 1a). We note that the perovskite structure of TbMnO₃ is distinct from that of the antiferromagnetic (AF) ferroelectric hexagonal rare-earth manganites (for example, YMnO₃) in which magnetoelectric coupling has been well studied^{5–9}. Furthermore, the perovskite TbMnO₃ does not contain 6*s* lone pairs, which produce a polar structure in magnetic ferroelectric perovskites BiMnO₃ (refs 11–13) and BiFeO₃ (ref. 14). The electronic configuration of the Mn³⁺ site in TbMnO₃ is identical with that in the parent compound of colossal magnetoresistive manganites, LaMnO₃, having the $t_{2g}^3 e_g^1$ configuration. In LaMnO₃, staggered orbital order of the $d_{3x^2-r^2}/d_{3y^2-r^2}$ type is responsible for the layered (A-type) AF order. In contrast, the spin structure in TbMnO₃ is a sinusoidal AF ordering of the Mn³⁺ moments that takes place at



Figure 1 Appearance of ferroelectricity below a lock-in transition temperature in TbMnO₃. **a**, Rough sketches of crystal structure at room temperature (top) and spatial variation along the *b* axis of Mn magnetic moment and atomic displacement ($\Delta z//c$) below T_N (lower). Orange arrows denote Mn magnetic moments below T_N . **b**–**e**, Temperature profiles of magnetization and specific heat divided by temperature $C/T(\mathbf{b})$, wavenumber of lattice modulation $k_1(\mathbf{c})$, dielectric constant ϵ at 10 kHz (**d**), and electric polarization P along the principal axes in single crystals of TbMnO₃ (**e**). The error bars of k_1 denote the distribution of k related to the correlation length.

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 $T_{\rm N} \approx 41 \,\mathrm{K}$ with a wave vector $\mathbf{q} = (0, k_s, 1)$ in the *Pbnm* orthorhombic cell¹⁶. A Mn³⁺ moment oriented along the *b* axis has been proposed¹⁶. The wavenumber k_s is incommensurate (~0.295) at $T_{\rm N}$ and decreases with decreasing temperature, becoming nearly constant (0.28) below ~30 K. A rough sketch of the magnetic order on Mn moments is illustrated in Fig. 1a. Recently, the emergence of long-period sinusoidal AF order in TbMnO₃ has been explained in terms of spin frustration caused by the combination of a GdFeO₃-type distortion and staggered orbital order¹⁷.

We show in Fig. 1b temperature (*T*) profiles of magnetization (*M*) at 0.5 T, and of specific heat divided by temperature (*C*/*T*), for a single crystal of TbMnO₃. Both *M* and *C*/*T* exhibit three anomalies at \sim 7 K, \sim 27 K and \sim 42 K. Taking account of results of a former neutron diffraction study¹⁶, the anomalies at \sim 7 K and \sim 42 K are attributed to the magnetic ordering of the Tb³⁺ moments and the sine-wave ordering of the Mn³⁺ moments, respectively. The anomaly at 27 K seems to be related to the incommensurate–commensurate (or lock-in) transition where the magnetic modulation wave vector **k** is locked at a constant value.

X-ray diffraction measurements revealed that the modulated magnetic ordered phase is accompanied by magnetoelastically induced lattice modulation¹⁷. Figure 1c displays the T dependence of wavenumber k_1 and normalized intensity of a superlattice reflection at $(0,k_l,3)$ attributed to the lattice modulation. The superlattice reflections at the wave vector $(0, k_b l)$ with l an integer, appear below $T_N \approx 41$ K ($k_l \approx 0.57$ at T_N). The intensity of the superlattice reflections increases down to the lock-in transition temperature ($T_{\text{lock}} \approx 27 \text{ K}$), but is suppressed again below T_{lock} . Another notable feature is the T dependence of the modulation wavenumber k_b as clarified in Fig. 1c. Upon cooling from T_N down to T_{lock} , k_l decreases and then becomes nearly constant ($k_l \approx 0.55$) below T_{lock} . The value of the T-dependent k_l is almost twice as large as that of the magnetic wavenumber k_s (ref. 16). It is well known that the crystallographic deformations upon magnetic ordering are caused by the magnetic atoms increasing their exchange interaction energy by shifting their positions; that is, exchange striction^{18,19}. By analogy with the lattice modulations observed in some rare-earth metals such as Tm with sinusoidal order^{20,21}, the observed superlattice reflections due to atomic displacement can be regarded as a second harmonic that is magnetoelastically induced by sinusoidal AF order.

In materials with incommensurate phases²², such as the family of A₂BX₄ compounds (for example, Rb₂ZnCl₄, (NH₄)₂BeF₄ and K₂SeO₄), the lattice modulation is generally connected with a spatially varying electric polarization. With further decreasing temperature, these materials exhibit a first-order phase transition (lock-in transition) to a ferroelectric phase with finite spontaneous polarization. Taking account of these observations, we could expect to find ferroelectricity in TbMnO3 associated with the incommensurate phase at the lock-in transition. Thus, we performed measurements of the dielectric constant, ϵ , and the electric polarization, P, along the *a*, *b* and *c* axes. Figure 1d displays *T* profiles of ϵ at 10 kHz. While ϵ_b (electric field E parallel to the b axis; E||b| shows merely weak T dependence, several anomalies are observed in $\epsilon_a(E||a)$ and ϵ_c (E||c). ϵ_a increases below $T_N \approx 41$ K towards low temperature, and becomes nearly constant below T_{lock} . The most striking feature is observed in ϵ_c . A pronounced λ -type peak of ϵ_c is found at T_{lock} , although no distinct anomaly is observed at T_N. The sharp peak structure in ϵ_c at T_{lock} is suggestive of the occurrence of the ferroelectric phase transition. To confirm the ferroelectric activity, we show in Fig. 1e the temperature variation of P. A finite spontaneous polarization appears below T_{lock} along the c axis. P_c (P||c) at ~10 K is about 8 × 10⁻⁴ C m⁻². We also confirmed that P_c can be reversed by the d.c. electric field. Thus, TbMnO3 becomes ferroelectric below T_{lock} .

The spontaneous polarization of TbMnO₃ is rather small compared with that of conventional ferroelectrics (for example, $\sim 2.6 \times 10^{-2} \text{ Cm}^{-2}$ at 296 K in BaTiO₃), but is comparable to that of the so-called improper ferroelectrics²³ (for example, $\sim 1.2 \times 10^{-3} \text{ Cm}^{-2}$ at 153 K in Rb₂ZnCl₄ (ref. 24) and $\sim 5.6 \times 10^{-4} \text{ Cm}^{-2}$ at 77 K in K₂SeO₄ (ref. 25)). In improper ferroelectrics, the primary order parameter represents the lattice distortion mode having a nonzero wave vector (that is $k_l \neq 0$), and the spontaneous polarization appears as a secondary order parameter induced by the lattice distortion. This may also be the case for TbMnO₃. An uncompensated antidipole structure is possible in





polarization along the *c* and *a* axes (**c** and **d**), respectively, at various magnetic fields in single crystals of TbMnO₃. Magnetic fields are applied along the *b* axis.

improper ferroelectrics, which was specified by the term ferrielectricity²⁶ because of its analogy with ferrimagnetism. Indeed, the intensity comparison of the superlattice reflections at $(0,k_l,0)$ and $(0,k_l,2)$ measured by the X-ray diffraction suggests that the atomic displacement in the ferroelectric phase has a component along the *c* axis, that is, the direction of the spontaneous polarization (Fig. 1a). In addition, weak intensities of the superlattice reflections (five orders of magnitude smaller than the Bragg peak) and the lack of their enhancement on tuning the X-ray photon energy at Mn *K*- and Tb *L*-absorption edges suggest the lattice distortion is due to the displacement of O ions.

As the lattice modulation in TbMnO₃ is accompanied by magnetic order, we may expect coupling between magnetization and polarization. We have investigated the effect of magnetic field on ϵ and *P*. In the experiments, magnetic fields (*B*) were applied along the *b* axis, that is, the direction of the modulation wave vector and the Mn³⁺ magnetic moments. Figure 2a and b shows the *T* profiles of ϵ along the *c* and *a* axes, respectively, at various magnetic fields. No distinct change of ϵ for either direction was observed at temperatures above T_{lock} . Although the sharp peak at T_{lock} in ϵ_c is not sensitive to the application of magnetic fields, another peak feature at $T_{\text{flop}}(B)$ shows up in ϵ_c below T_{lock} above 5 T. The magnetic-field-induced sharp maximum is shifted towards higher *T* with

increasing magnetic field from $T_{\rm flop}(5 \text{ T}) \approx 8 \text{ K}$ to $T_{\rm flop}(9 \text{ T}) \approx 20 \text{ K}$. The component ϵ_a also shows a remarkable magnetic field effect. By applying magnetic fields above 5 T, ϵ_a exhibits a peak structure at $T_{\rm flop}$, and shows a thermal hysteresis at a temperature between $T_{\rm flop}$ and $T_{\rm lock}$. We also measured the ϵ_b in magnetic fields, but no substantial magnetic field effect was observed up to 9 T.

Measurements of *P* in magnetic fields have revealed the origin of the magnetic-field-induced anomaly in ϵ . We show in Fig. 2c and d the *T* variation of P_c and P_a , respectively, at various magnetic fields. By applying a magnetic field above ~5 T, the spontaneous polarization in P_c is considerably suppressed below $T_{\rm flop}$. With increasing magnetic field, the temperature region with finite P_c shrinks, corresponding to the shift of $T_{\rm flop}$. By contrast, the application of a magnetic field induces a finite P_a (Fig. 2d). The onset temperature of a finite P_a coincides well with $T_{\rm flop}$, and increases in accord with the increase of $T_{\rm flop}$ by applying a magnetic field. These results show that the spontaneous polarization is switched ('flopped') from the direction along the *c* axis to the direction along the *a* axis at $T_{\rm flop}$, and the strong magnetic field dependence of $T_{\rm flop}$ causes the notable magnetic field effect on ϵ and *P*.

We display in Fig. 3 the magnetic field dependence of the change in ϵ ($\Delta\epsilon(B)/\epsilon(0) = [\epsilon(B) - \epsilon(0)]/\epsilon(0)$), *P* and magnetization at selected temperatures. As shown in Fig. 3a and b, a remarkable



Figure 3 Magnetocapacitance and magnetoelectric effects in TbMnO₃. **a**–**e**, Magnetic-field-induced change in dielectric constant (**a** and **b**), electric polarization along the *c* and *a* axes, respectively (**c** and **d**), and magnetization at selected temperatures (**e**). Magnetic fields are applied along the *b* axis. The magnetic field dependence of electric

polarization was obtained by the measurements of magnetoelectric current, which were performed after poling crystals. The data of **d** were collected after the magnetic field cooling. The numbers in **d** denote the order of measurements at 9 K.

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Figure 4 Temperature versus magnetic field phase diagram for TbMnO₃ for magnetic field applied along the *b* axis. Open and filled symbols represent the data points in the cooling (or magnetic-field increasing) and warming (or magnetic-field decreasing) runs, respectively. T_{N} (Mn) (determined from the dielectric anomaly in ϵ_{a}) and T_{N} (Tb) (determined from the anomaly in the *M*–*T* curve) indicate the antiferromagnetic ordering

temperature of the Mn³⁺ and Tb³⁺ moments, respectively. T_{lock} and T_{flop} , which were determined from the dielectric anomaly, denote the temperatures of incommensurate– commensurate (or lock-in) transition and electric polarization flop, respectively. Triangles indicate the points where the magnetization curves show steep steps. The shaded areas show magnetic field hysteresis regions.

peak structure with hysteresis is observed at almost the same magnetic field for both $\Delta \epsilon_c(B)/\epsilon_c(0)$ and $\Delta \epsilon_a(B)/\epsilon_a(0)$ at their respective temperatures. The peak position $B_{\rm flop}(T)$ is shifted towards higher magnetic field with increasing temperature. The maximum value of the magnetocapacitance as defined by $\Delta \epsilon(B_{\rm flop})/\epsilon(0)$ reaches values as large as ~10% at 12 K. The divergent increase of ϵ at $B_{\rm flop}$ has an intimate connection to the magnetic-field-induced electric polarization flop, which is confirmed by the magnetic field dependence of P (Fig. 3c and d). The spontaneous polarization along the c axis is suddenly suppressed at $B_{\rm flop}$, whereas that along the a axis appears at $B_{\rm flop}$ with increasing magnetic fields. The data in Fig. 3c and d can be regarded as a novel magnetoelectric effect, where the gigantic change in $P(\Delta P(B) \approx 6 \times 10^{-4} \text{ Cm}^{-2})$ is caused by the magnetic-field-induced electric polarization flop with the nature of first-order phase transition.

To sum up the present study, we have determined the electric and magnetic phase diagram of TbMnO₃ from dielectric and magnetic anomalies for the direction of magnetic field parallel to the b axis (Fig. 4). The magnetic phase boundaries denoted by triangles are determined from steep steps in magnetization curves (Fig. 3e). This clearly indicates that $T_{\text{flop}}(B)$ is in good agreement with a magnetic phase boundary. Similar metamagnetic transitions also take place in a rare-earth orthoferrite, TbFeO₃, and this has been interpreted as the magnetic reversal of Ising Tb³⁺ moments^{27,28}. Furthermore, the Tb^{3+} spin reversal gives rise to the spin reorientation in Fe sites. By analogy, it is likely in the magnetic-field-induced electric polarization flop of $TbMnO_3$ that Tb^{3+} moment reversal, accompanied by the Mn spin reorientation, changes the exchange interaction energy and then brings about the lattice modulation owing to a finite spontaneous polarization along the *a* axis. The giant magnetocapacitance and magnetoelectric effects as observed in the present frustrated-spin system may provide intriguing opportunities to explore a new type of magnetoelectric functionality. \square

Methods

Single crystals of TbMnO₃ were grown in a flow of Ar by the floating zone method. The crystals were oriented using Laue X-ray diffraction patterns, and cut into thin plates (typical dimensions, $\sim 3 \times 3 \times 1 \text{ mm}^3$) with the widest faces perpendicular to the crystallographic principal axes. The magnetization and specific heat for the crystals were measured with a commercial magnetometer and a relaxation technique, respectively. Single-crystal diffraction measurements were performed at BL-4C of the Photon Factory, KEK, in Tsukuba. X-rays with photon energy of 13 keV were used for the non-resonant X-ray scattering, and that near the Mn *K*- and Tb *L*-absorption edges for the resonant one. The dielectric constant was measured at 10 kHz using an LCR meter. The *T* dependence of electric polarization was obtained by measurements of pyroelectric current. The sample was cooled down while applying a poling field ($\sim 150 \text{ kV m}^{-1}$). At the lowest *T*, the poling electric field was removed. Then, the sample was heated at a constant rate (5 K min⁻¹), and the pyroelectric current was measured. For these electric measurements, silver electrodes were evaporated onto the widest faces of the sample.

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