# Supplemental Information for "Symmetry crossover in layered $MPS_3$ complexes (M = Mn, Fe, Ni) via near-field infrared spectroscopy"

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#### Far field infrared absorption, Raman scattering, and near field infrared

Figure S1 displays the traditional infrared absorption and Raman scattering response of MnPS<sub>3</sub>, FePS<sub>3</sub>, and NiPS<sub>3</sub> single crystals at room temperature. This figure also shows (in the bottom set of panels) the corresponding near-field spectra, both amplitude and phase, for each system. The signal-to-noise ratio in the near-field spectrum is much lower than what one achieves with standard far-field infrared techniques. This is because we are dealing with exfoliated flakes that are extremely small in size with little to no optical density. Increasing the number of scans can provide some relief because the signal-to-noise ratio goes as the square root of the number of scans. Additional scans will not, however, make a small spectral feature become larger. On the other hand, this spatially-resolved tip-based nature of the technique is a tremendous advantage when measuring exfoliated sheets.



FIG. S1. Top panels (a-c) display traditional infrared absorption and Raman scattering spectra for single crystals of  $MnPS_3$ ,  $FePS_3$ , and  $NiPS_3$  at 300 K. The bottom panels show the near-field infrared amplitude and phase of the corresponding systems.

## Vibrational mode assignments

Table S1 summarizes the mode assignments for the  $MPS_3$  (M=Mn, Fe, Ni) family of materials across the currently available near field frequency window (330-700 cm<sup>-1</sup>). These assignments are made based upon the data in Fig. S1, comparison with literature results, [S1–S4] and our complementary lattice dynamics calculations.

Material	Infrared	Raman	Near-field* Assignments	
$\mathbf{MnPS}_{3}$	-	385	365	$ u(\mathrm{PS}_3)$
	452	-	450	$T'_{xy}(\mathrm{PS}_3) + \nu(\mathrm{P-P})$
	573	567, 581	567,609	$ u(\mathrm{PS}_3)$
$\mathbf{FePS}_3$	-	378	357	$ u(\mathrm{PS}_3)$
	448	-	446	$T'_{xy}(\mathrm{PS}_3) + \nu(\mathrm{P-P})$
	572	580	580,611	$ u(\mathrm{PS}_3)$
$\mathbf{NiPS}_3$	-	350,  385	-	$ u(\mathrm{PS}_3)$
	-	435	-	$T'_{xy}(\mathrm{PS}_3) + \nu(\mathrm{P-P})$
	548, 575	559, 586	584	$ u(\mathrm{PS}_3)$

TABLE S1. Vibrational mode assignments of  $MnPS_3$ ,  $FePS_3$ , and  $NiPS_3$  single crystals. All values are in cm<sup>-1</sup>.[S1–S4]

 $\nu$  = symmetric stretch, T' = translational motion, \* corresponds to maxima in the phase spectra.

## The $B_u$ symmetry vibrational mode is an incisive probe of local symmetry

Figure S2 summarizes the behavior of the  $B_u$  symmetry vibrational mode in the  $MPS_3$ (M = Mn, Fe, Ni) family of materials. We measured these single crystals in two different ways: using traditional far infrared absorption spectroscopy and via synchrotron-based near field nano-spectroscopy. Both reveal that the Ni compound is special. Figure S2(a) shows that the  $B_u$  mode is present in the MnPS<sub>3</sub> and FePS<sub>3</sub> systems. It is not present in NiPS<sub>3</sub>. This is consistent with the near-field studies shown in Fig. S2(b) where the  $B_u$  symmetry vibrational mode is again missing. These systems have been traditionally assigned to the C2/m space group in their bulk form.[S1, S5, S6] The spectra of MnPS<sub>3</sub> and FePS<sub>3</sub> are consistent with this assignment. On the other hand, the lack of a  $B_u$  vibrational mode in NiPS<sub>3</sub> unequivocally demonstrates that the three-fold rotational axis has been restored. This is consistent with  $P\bar{3}1m$ . Figure S2(c) shows schematically how the presence (or absence) of the  $B_u$  vibrational mode is related to the absence (or presence) of the C<sub>3</sub> rotation.



FIG. S2. (a) Close-up view of the  $B_u$  vibrational mode in single crystal MnPS<sub>3</sub>, FePS<sub>3</sub>, and NiPS<sub>3</sub> via traditional far infrared spectroscopy at room temperature. This structure is absent in the Ni compound. (b) Close-up view of the  $B_u$  mode in the near field infrared response of the same set of materials. Clearly, the  $B_u$  mode is absent in NiPS<sub>3</sub> in both far- and near-field studies - even though it is amplified in the near field spectrum of MnPS<sub>3</sub> and FePS<sub>3</sub>. (c) Schematic view of how loss of this particular mode signifies the restoration of a three-fold rotational axis and therefore a symmetry increase to space group to  $P\bar{3}1m$ . Conversely, the presence of the  $B_u$  symmetry feature is microscopic evidence that the system exists in the C2/m space group.

### Comparison of local structure in this family of materials

We also summarize the local structure of  $MnPS_3$ ,  $FePS_3$ , and  $NiPS_3$  [Table S2]. [S6, S7] Bond length and angle trends are discussed in the main text along with how these trends dovetail with the crystal symmetry.

### Modeling frequency vs. layer number trends

Thinning the crystals to the monolayer is akin to confinement in that the overall length scale is being reduced. To make this more apparent, we fit the layer (thickness) dependent phonon frequencies with a confinement model  $\omega = \omega_{bulk}(\frac{q}{r^2})$  in Fig. 3 (c) of the main text. Here,  $\omega$  is the frequency of the  $B_u$  mode of the sheet,  $\omega_{bulk}$  is the bulk frequency of the same  $B_u$  symmetry mode, r is the layer number (which is directly proportional to sheet thickness), and q is a scaling factor that represents the sensitivity of a particular phonon to confinement.

Bond/angle type	$\mathbf{MnPS}_3$	Number	$\mathbf{FePS}_3$	Number	$\mathbf{NiPS}_3$	Number
P-S bond (Å)	2.0735	4	2.287	4	1.9905	4
	2.0765	2	2.0323	2	1.9935	2
P-P bond (Å)	2.2154	1	2.2099	1	2.1454	1
M-S bond (Å)	2.5887	3	2.5498	3	2.4993	3
	2.5942	3	2.5567	3	2.4999	3
	2.5818	4	2.5432	4	2.4997	4
S-P-S angle (°)	114.640	2	114.417	2	113.812	2
	114.290	4	114.083	4	113.988	4
S- $\overline{M}$ -S angle (°)	94.302	2	95.148	2	95.316	2
	94.150	4	94.988	4	94.972	4

TABLE S2. Bond lengths and angles for the  $MPS_3$  (M = Mn, Fe, Ni) family of materials at 300 K.[S6, S7]

Reasonable fits are obtained for each data set by fixing  $\omega_{bulk}$  as the measured bulk value. This leaves q as the only fitting parameter which is obviously material dependent. We find q = 73 and 53 for MnPS<sub>3</sub> and FePS<sub>3</sub>, respectively. The higher value for MnPS<sub>3</sub> indicates that the system is overall stiffer. We find that the fit quality in Fig. 3(c) is better for MnPS<sub>3</sub> than for the Fe analog.

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