Supplemental Material for

Lattice dynamics and structural transition of the hyperhoneycomb iridate β-Li_2IrO_3 investigated by high-pressure Raman scattering

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This Supplemental Material present the full technical details of single crystal growth and characterization in Section S1, Raman measurements to characterize beam-heating in Section S2, Raman measurements in Section S3, followed by details of structural relaxations in Section S4 and calculations of phonon frequencies in Section S5, and computational details in Section S6. Tables provide relative fitted peak positions from the beam-heating measurements and DFT/DMFT-optimized crystal structures at both ambient and high pressure.

S1. SINGLE CRYSTAL GROWTH AND CHARACTERIZATION

High-quality single crystals were synthesized by a flux-method explained elsewhere.[1] The crystallographic axes of the single crystals were determined by single crystal x-ray diffraction and polarization-resolved Raman measurements by means of the Raman selection rules.[2] The crystal morphology was plate-like with the ***c***-axis perpendicular to the plane, similar to its structural polytype γ-Li_2IrO_3.[3] [see Fig. S1a) for a microscopic image]. Typical crystal dimensions used for Raman experiments are about 30μm $\times$ 20μm $\times$ 15μm.

S2. RAMAN MEASUREMENTS TO CHARACTERIZE BEAM-HEATING

To estimate extrinsic shifts of phonon peaks in the room temperature measurements by a local beam-heating of crystals, we carefully determined the maximised Raman signal to noise ratio that we could get while we avoided the beam-heating for all experimental setting that we used: 50x, 20x without the Diamond Anvil Cell (DAC) and 20x with the DAC in both green and red lasers. Figure S1a) shows a typical morphology of the crystal used in the experiment with the corresponding unit cell with the iridium network in Fig. S1b).

Figure S2 summarizes the softening of selected phonon peaks with an elevating Raman laser power: we chose $A_g(1)$, $A_g(4)$, $A_g(6)$ ($A_g(4)$, $A_g(6)$) with the red laser (with the green laser) as they are stronger than other peaks. By fitting normalised fitted peak positions [explicitly given in Table I from a pseudo Voigt fit, a combined function with Lorentzian and Gaussian profiles], we found that the threshold Raman power (not inducing the beam-heating effect) is about 0.7 mW without the DAC [Fig. S2a)] and about 2 mW with the DAC [Fig. S2b)]: due to a bigger attenuation of light intensity by the DAC, the threshold laser power increases with the DAC [as compared in Figs. S2a) and S2b)]. To avoid the beam-heating, these threshold Raman laser powers were used for all relevant Raman measurements presented in this paper.

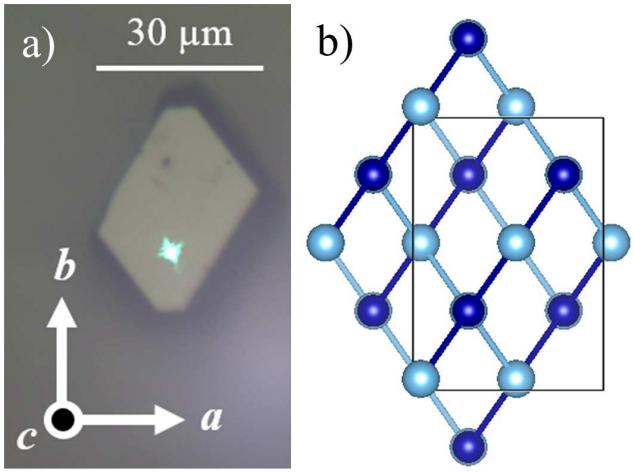

FIG. S1. (color online) a) A microscopic image of a representative β-Li_2IrO_3 crystal with a green laser at ambient pressure without the DAC, focused with a 50x microscope. b) The same orientation of the unit cell with Ir ions (blue and dark blue balls). Oxygen and lithium ions are not displayed for a simplicity.

S3. RAMAN MEASUREMENTS

Raman experiments with the green laser were performed with the 514 nm excitation line of an argon/krypton laser using a JobinYvon T64000 spectrometer with an energy resolution of ~ 2.4 cm^{-1} (measured by a neon lamp). The measurements with the red laser

TABLE I. Fitted peak positions with the red and green Raman data upon the laser power. A lens of 20x was used with the DAC and a 50x lens was utilized without the DAC. The R (G) symbol in the first row means the red (green) laser. The unit of frequencies is cm^{-1}. The typical fitting error bars are in the two decimal places.

Power	A_g(4)G	A_g(6)G	Power	A_g(1)R	A_g(4)R	A_g(6)R	Power (DAC)	A_g(4)G	Power (DAC)	A_g(4)R
0.40 mW	518.540	597.422	2.65 mW	181.899	514.702	617.211	0.52 mW	532.072	0.675 mW	517.569
0.50 mW	518.579	596.928	2.52 mW	181.883	515.624	618.024	0.68 mW	532.866	0.682 mW	518.797
0.60 mW	517.970	597.131	1.76 mW	182.185	517.382	620.397	0.92 mW	532.382	1.2 mW	517.181
0.70 mW	518.438	596.944	1.48 mW	181.674	517.299	620.999	1.34 mW	532.846	1.62 mW	518.112
0.80 mW	517.614	596.153	1.25 mW	182.880	517.749		1.7 mW	532.451	2 mW	518.128
0.91 mW	517.452	596.514	1.00 mW	182.543	518.406	622.438	1.98 mW	530.899	2.45 mW	517.061
1.01 mW	516.893	595.611	0.71 mW	182.964	520.274		2.29 mW	530.606	2.87 mW	515.678
1.17 mW	515.929	595.570	0.50 mW	183.748	519.837	623.417	3.16 mW	530.075	3.63 mW	515.365
1.53 mW	514.842	594.527	0.33 mW	183.679	519.715	623.918	3.82 mW	530.721		
2.09 mW	512.637	592.528	0.23 mW	183.258	519.852	623.999	4.7 mW	530.383		
2.50 mW	511.614	591.268								

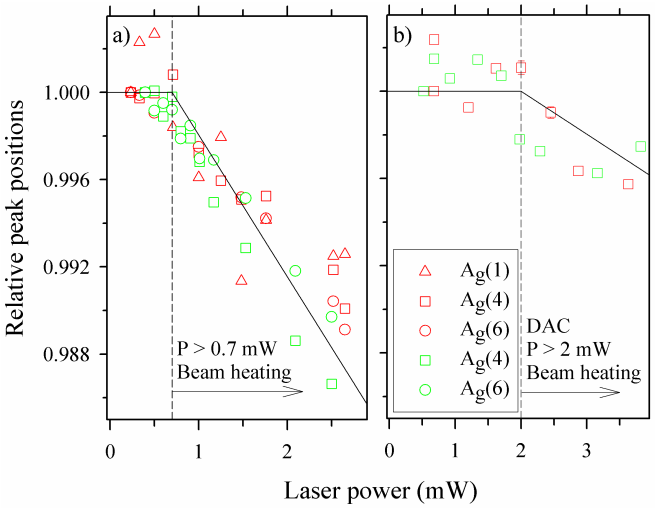

FIG. S2. (color online) Normalized positions of selected Raman peaks in terms of laser powers at ambient pressure a) without the DAC and b) with the DAC. Green (red) symbols are phonon peaks obtained by using the green (red) Raman laser. Two vertically dashed black lines indicate the estimated (based on the fits with solid black lines) strongest laser power at about 0.7 mW (2 mW) without (with) the DAC, which do not induce an artificial beam-heating effect.

used the 632.8 nm line of a HeNe mixed gas laser and a Labram (Horiba Jobin-Yvon) single-grating spectrometer.[4,5]

First, we checked the Raman spectra at ambient pressure with various polarizations: measurements did not show any meaningful differences in the Raman spectra, indicating homogeneous sample quality and compositions for good crystals.

We also obtained the Raman data with the rotated crystal with various in-plane angles along the perpendicular direction (along the $\boldsymbol{c}$-axis) of the ($\boldsymbol{ab}$)-plane-oriented crystal [see Fig. S1a)]: we confirmed that small misalignment within the plane ($\Delta\theta \lesssim 15°$) did not give very noticeable change in the spectra, ensuring reliable Raman spectra collected at even high pressure: note that a negligible misorientation of the in-plane orientation ($\lesssim 5°$) was unavoidable during high-pressure Raman measurements.

The diameter of the beam was typically $\lesssim 5$ μm at ambient pressure without the DAC, measured by a varying size of circles of gold deposited to the Al_2O_3 substrate using a sharp contrast of Raman signals from gold[6](a broad continuous intensity) and Al_2O_3[7] (a set of sharp phonon peaks). Without the DAC, we confirmed that the effective size of Raman light is approximately similar to the size of circular light observed under the microscope. Based on this, we estimated the beam size inside the DAC, to be $\lesssim 30$ μm (estimated only with the red Raman light as green Raman lights give a much smaller beam size even with the DAC).

Systematic and accurate measurements were pursued by controlling various experimental conditions. For instance, the identical microscopic lens were used for all measurements: Nikon 50x/0.45 Super Long Working Distance (SLWD) and 20x/0.35 SLWD lens (the largest magnification lens available to us to be compatible with our diamond anvil cell) to use intentionally the same attenuation rate of light. It was because different types of lens would have different attenuation rates for the given light, so the beam-heating rate, which was obtained from the analysis shown in Fig. S2, would be modified accordingly.

With the green Raman light (not necessarily with the red Raman light), the continuous flow of Ar-gas has been implemented to effectively suppress Raman signals from the vibrational air scattering, mostly below about 150 cm^{-1}, which was crucial to reliably identify and trace phonon peaks at the low-energy transfer especially at high pressures. In green and red Raman experiments,

different single crystals were used with a similar sample quality.

Moreover, Raman measurements on other beam positions at both ambient and finite pressures with the DAC were tested, finding only a mere change in the background signal. Small linear background signals (mostly coming from the DAC) were subtracted for some high-pressure data when necessary for a better representation. A small variation of the background signal at different pressure and polarization seems to be originated from a slight redistribution of the medium liquid (see Section S2B for details) when the new pressure was applied and/or the shape and size of the Raman light was changed depending on the incident polarization of the light. All Raman measurements were made with a high-resolution (1800 grooves/mm) setting to measure the Raman spectra more precisely.

A. Polarized measurements with green laser (514.5 nm)

We should mention that there are some ambiguities to identify weak and overlapped Raman peaks at high pressures. For example, a lesser number of Raman peaks has been experimentally measured reliably, compared with 18 Raman-active phonon modes obtained from the calculation [compare Table I and II in the main text] possibly due to their weak Raman signals. In fact, we observed very weak peak-like and shoulder features in the Raman spectra at higher pressures, but their tiny intensities for the whole pressure range explored did not allow us to do a reliable fit, so they were not marked in Fig. 2 and Table II in the main text. Moreover, in Fig. 2 in the main text, it was nearly impossible to do a reliable fit with collected Raman data at P = 4.53 and 5 GPa due to weak intensities of new Raman peaks with an increased signal from the DAC, thus we best estimated peak positions at P = 4.53 and 5 GPa reversely from peak positions reliably identified from P = 6.3 and 7.62 GPa, whose peaks were much better defined. Furthermore, we cannot completely rule out a possibility of mixing of sample peaks with the DAC peaks: i.e., a peak-like signal between 220 cm^{-1} and 250 cm^{-1} for a $\bar{\boldsymbol{c}}(\boldsymbol{bb})\boldsymbol{c}$ polarization in Fig. 2b) in the main text already present even at 2.4 GPa below the transition.

Asymmetric profiles of some peaks (i.e., peaks at lower wavenumbers) could be also from the combination of multiple peaks nearby, or from the coupling with electronic response with the Ir $J_{\rm eff} = 1/2$ local moments similarly seen in Sr_2IrO_4.[4] For testing the latter case, the fitting with a Fano asymmetry profile[8] was attempted, but did not give any noticeable trend in the fitted parameters (i.e., linewidths). If the same physics should apply to this compound, the absence of this coupling is probably due to weak Raman intensities coexistent with increased background signals when equipped with the DAC.

B. High-pressure measurements

High-pressure Raman measurements were performed with the DAC. Diamond anvils had culet diameters of 0.4 mm and were of the ultra-low luminescence type. The stainless steel gaskets were preindented to 100 μm thickness and a hole of 175 μm diameter was drilled into each gasket by spark erosion. The hole was designed to ensure enough space for the thick β-Li_2IrO_3 single crystals. Several attempts with thinner gaskets failed by breaking samples at intermediate pressures during Raman measurements.

High-pressure Raman data showed weaker signals when equipped with the DAC possibly due to the strong background signal from the DAC [see Fig. 1 in the main text], an enhanced light attenuation by the DAC and a less focusing light due to a decreased magnification of available lens (from 50x to 20x): our high-pressure setup with the DAC was not compatible with the focal length of the 50x lens, whereas Raman signals without the DAC allowed a larger magnification lens (50x). The latest factor increased a typical measurement time ($\gtrapprox$ 10 hours) for a single Raman data [i.e., Fig. 2 in the main text] at one pressure and polarization to maximize the signal to noise ratio.

Porto's notation[9] was utilised to describe the configuration of the Raman scattering experiment (in a backscattering geometry with the light propagating along the crystalline $\boldsymbol{c}$-axis). It expresses the orientation of the crystal with respect to the polarization of the Raman laser in both exciting and analysing directions, in a form of $k_i(E_iE_s)k_s$, where k_i (k_s) is the direction of incident (scattered) light and E_i (E_s) is the polarization of incident (scattered) light, respectively.

At high pressure, the crystallographic axes of the monoclinic structure are different from those in ambient orthorhombic structure since the $\boldsymbol{c}$-axis is no longer parallel to the the vertical axis of the laboratory frame, but tilted by 16.777° from the normal direction.[10] However, we did not observe any significant difference in the measured spectra for $\bar{\boldsymbol{c}}(\boldsymbol{ab})\boldsymbol{c}$, $\bar{\boldsymbol{c}}(\boldsymbol{ba})\boldsymbol{c}$ polarizations as shown in Fig. 2 in the main text, indicating an insensitivity of this tilted angle of the $\boldsymbol{c}$-axis in our measurements. This is consistent with our previous characterization measurements, which only showed some meaningful variations in the Raman spectra when the crystal was rotated by $\sim$ 15° in the ($\boldsymbol{ab}$)-plane at ambient pressure [as explained in S3]: this makes our polarization analysis reliable even at high pressures.

By symmetry analysis, Raman tensors[11] of high-pressure monoclinic structure[10] are given as

$$I_{001}(\mathrm{A_g}) = \begin{pmatrix} \mathrm{A} & \mathrm{D} & \\ \mathrm{D} & \mathrm{B} & \\ & & \mathrm{C} \end{pmatrix}$$

$$I_{001}(B_g) = \begin{pmatrix} & & \mathrm{E} \\ & & \mathrm{F} \\ \mathrm{E} & \mathrm{F} & \end{pmatrix}, \quad (1)$$

where A, B, C, D, E and F are Raman intensity components and a subscript is the direction of the propagating light (the monoclinic $\boldsymbol{c}$-axis).

S4. STRUCTURAL RELAXATIONS

At ambient pressure, both SOC and U were essential (as explained in the main text) to stabilize the experimental structure. Otherwise, the calculations in the absence of either U or SOC found that an initial orthorhombic structure (close to the ideal hyperhoneycomb structure) became unstable and evolved into a new type of Ir-dimerized orthorhombic crystal structure at ambient pressure, destroying the Ir $J_{\mathrm{eff}} = 1/2$ local moments [dimerized along the $\boldsymbol{c}$-axis in Fig. 4a) in the main text]. On the other hand, when we kept the converged electronic structure with the $J_{\mathrm{eff}} = 1/2$ moments and pressurized the unit cell (i.e., optimizing the cell parameters and internal coordinates with a smaller fixed volume), the orthorhombic phase without the Ir dimerization was maintained up to 10 GPa[12] as a local minima state.

S5. CALCULATIONS OF PHONON FREQUENCIES

A. DFT+SOC+U results

In this section, we discuss our theoretical attempts to understand origins of their mismatches in calculations and experiments. Comparing in Table I and Table II in the main text, at ambient pressure, the calculated phonon frequencies agree well with the observed phonons fitted from the Raman data except for two peaks in the spectral range between 600 and 680 cm^{-1} in the $\mathrm{A_g}$ channel. In particular, the frequency difference between the highest measured and calculated $\mathrm{A_g}$ modes is about 100 cm^{-1}, which is consistently reproduced by alternative DFT+SOC+U calculation with WIEN2K code. On the other hand, at high pressure, the calculated phonon energies agree better with the measured peaks fitted from the Raman data in Figs. 2a-d) in the main text (compare Table I and II in the main text): the agreement is slightly worse at a lower energy, possibly due to less important (but non-negligible) roles of U and SOC in the dimerized structure.

We should also point out that although the overall calculating phonon frequencies match better at high pressure, a marginal mismatch of frequencies between the data and calculation is also attributed to the difference of pressures used for the comparison: a higher pressure (7.62 GPa) of the experimental data than the pressure used in the *ab-initio* calculations at (presumably) 4 GPa.[10]

B. Dynamical mean-field results on the highest A_{g} mode at ambient pressure

To understand the origin of the largest discrepancy between DFT+SOC+U and experimental Raman data for the high energy $\mathrm{A_g}$ mode at ambient pressure, we also employed a method that can better describe strong correlation physics, in particular the non-perturbative nature of Mott insulator in the paramagnetic state; we present dynamical mean-field result on the highest A_{g} mode at ambient pressure.

For this calculation, we used the charge-self-consistent DFT+embedded dynamical mean-field theory (eDMFT) method,[13–15] (combined with WIEN2K[16]) including SOC to describe the *paramagnetic* Mott phase of the orthorhombic structure [see S6B for computational details]. The crystal structure optimized within eDMFT (at $T =$ 232 K) also showed a reasonable agreement with the experimental and DFT+SOC+U optimized structures [see Table II for details]. A finite displacement method for the highest A_{g} phonon mode was then used to draw the free energy versus the displacement curve for the calculation of the phonon frequency. As a result, the paramagnetic eDMFT predicts the frequency to be 556.6 cm^{-1}.

Interestingly, this result is very close to the WIEN2K magnetic DFT+U result, which is 550.9 cm^{-1}. This small ($\approx$ 5 cm^{-1}) difference in the frequency between the paramagnetic eDMFT and magnetic DFT+SOC+U results could indicate a negligible coupling between the magnetism and the lattice. The value obtained from the VASP method is about 587 cm^{-1}, hence the difference between two DFT codes (different for only $\sim$ 30 cm^{-1}) is larger than that between the DFT and eDMFT method. This is likely due to the basis set difference in the two DFT methods.

This is also consistent with our theoretical observation that a different magnetic order did not affect the relaxed crystal structure in the scheme of DFT+SOC+U once the Ir $J_{\mathrm{eff}} = 1/2$ state sets in.[12] However, it is also possible that the frustrated magnetism could play an important role in determining the highest A_{g} phonon energy (at about 587 $\mathrm{cm}^{-1}$from the VASP method) as this vibration is closely related to the local structure of Ir-O-Ir bond (as illustrated in Fig. 5d) in the main text), a key factor to determine exchange couplings.[12,17] Therefore, the origin of the mismatch of the highest A_{g} mode between the theory and experiment currently remains a topic for further investigation.

TABLE II. Experimental and optimized structural information of β-Li_2IrO_3 at ambient pressure. The space group is $Fddd$ (No. 70, origin choice 2), where the internal coordinates for each inequivalent site are $(1/8, 1/8, z)$ for Ir and Li1/2, $(x, 1/8, 1/8)$ for O1, and (x, y, z) for O2. In the DFT+SOC+U calculation, cell parameters ($\boldsymbol{a}$, $\boldsymbol{b}$ and $\boldsymbol{c}$) were allowed to change with the fixed volume, whereas in eDMFT fixed experimental cell parameters[1] were used. Ir-Ir and Ir-O bond lengths and Ir-O-Ir bond angles in each nearest neighbor bond are also given, where the Z- (X-) bonds denote Ir-Ir bonds parallel (not parallel) to along the $\boldsymbol{c}$-axis in Fig. 4a) in the main text. A zigzag-type antiferromagnetic and paramagnetic order was used for the DFT and eDMFT ($T = 232$ K) calculation, respectively. Both DFT calculations in the table used both SOC and U.

		Exp. (Ref. 1)	DFT (VASP)	DFT (WIEN2K)	eDMFT
	a	5.910	5.908	5.910	5.910
	b	8.456	8.440	8.456	8.456
	c (Å)	17.827	17.891	17.827	17.827
Ir ($16g$)	z	0.7085	0.7085	0.7096	0.7091
Li1 ($16g$)	z	0.0498	0.0448	0.0460	0.0459
Li2 ($16g$)	z	0.8695	0.8775	0.8783	0.8775
O1 ($16e$)	x	0.8572	0.8588	0.8614	0.8638
O2 ($32h$)	x	0.6311	0.6320	0.6294	0.6277
	y	0.3642	0.3654	0.3669	0.3666
	z	0.0383	0.0384	0.0389	0.0393
$d_{\mathrm{Ir-Ir}}$	Z	2.979	2.988	3.0203	3.000
(in Å)	X	2.973	2.973	2.9536	2.960
$d_{\mathrm{Ir-O}}$	Z	2.025	2.035	2.0573	2.059
(averaged)	X	2.025	2.029	2.0356	2.043
$\theta_{\mathrm{Ir-O-Ir}}$	Z	94.68	94.50	94.45	93.50
(degree)	X	94.43	94.23	93.02	92.86

S6. COMPUTATIONAL DETAILS

A. DFT+SOC+U calculations

We employed VASP to perform the electronic structure calculations, by using the projector-augmented wave basis set.[18,19] The same parameters for plane wave energy cutoff and k-point sampling used for the previous work[12] were chosen for the total energy and structural optimizations with experimental crystal structures at ambient[1] and high pressure.[10] The calculations with and without including atomic SOC, the DFT+U on-site Coulomb interaction,[20] and magnetism in the Ir d orbital were done. All of calculations shown in this paper were done with the value of $U = 2$ eV. We also checked that phonon spectra with $U = 2.5$ eV showed similar results compared to the $U = 2$ eV result (differences in frequencies smaller than 10 cm^{-1}).

In the phonon calculation, we noticed that the lightest Li ions did not contribute high-energy modes significantly although it is the lightest ions, probably due to the much weaker ionic bonding with other ions. This makes sense because Li ions in Lithium-ion battery cathode materials are considered to be more freely removed than other constituent ions, as experimentally observed in β-Li_2IrO_3 as well.[21]

TABLE III. Experimental and optimized lattice parameters and internal coordinates of β-Li_2IrO_3 with $C2/c$ (No. 15) space group symmetry at high pressure. In this calculation, lattice parameters of $\boldsymbol{a}$, $\boldsymbol{b}$, $\boldsymbol{c}$ and β were optimized in the DFT and DFT+SOC+U calculations with a fixed volume. Values of pressure measured in the experiment and DFT-estimation are shown in the top row. Ir-Ir and Ir-O bond lengths and Ir-O-Ir bond angles in each nearest neighbor bond are shown below.

		Exp. (Ref. 10)	DFT+SOC+U	DFT
	P (GPa)	4.4	5.0	5.4
	a	5.7930	5.7752	5.7485
	b	8.0824	8.0408	8.0319
	c (Å)	9.144	9.1951	9.2365
	β (degree)	106.777	106.263	106.016
Ir	x	0.4219	0.4235	0.4238
	y	0.3844	0.3877	0.3887
	z	0.0780	0.0772	0.0770
Li1	x	0.244	0.2434	0.2448
	y	0.632	0.6382	0.6401
	z	0.246	0.2436	0.2442
Li2	x	0.926	0.9270	0.9261
	y	0.625	0.6177	0.6165
	z	0.589	0.5936	0.5932
O1	x	0.7341	0.7320	0.7310
	y	0.3859	0.3895	0.3916
	z	0.2535	0.2542	0.2544
O2	x	0.9024	0.9024	0.9015
	y	0.3598	0.3596	0.3585
	z	0.5792	0.5811	0.5812
O3	x	0.4140	0.4118	0.4142
	y	0.3719	0.3625	0.3607
	z	0.5859	0.5870	0.5886
$d_{\mathrm{Ir-Ir}}$	dimer	2.6609	2.5999	2.5838
(in Å)	non-dimer	3.0136	3.0513	3.0697
$d_{\mathrm{Ir-O}}$	dimer	2.012	2.0098	2.0137
(averaged)	non-dimer	1.970	2.0235	2.0232
$\theta_{\mathrm{Ir-O-Ir}}$	dimer	84.3	80.6	79.8
(avg. deg.)	non-dimer	97.4	97.9	98.7

B. eDMFT calculations

A fully charge-self-consistent DMFT method,[14] implemented in DFT + Embedded DMFT (eDMFT) Functional code,[13] which is combined with WIEN2K code,[16] was employed for computations of electronic properties and optimizations of internal coordinates.[22] In DFT level

the Perdew-Burke-Ernzerhof (PBE) generalized gradient approximation (GGA) is employed,[23] and different choices of the DFT exchange-correlation functional may affect quantitative natures of the results presented here. 2000 k-points were used to sample the first Brillouin zone with $RK_{\max} = 8.0$. A force criterion of 10^{-4} Ry/Bohr was adopted for optimizations of internal coordinates. A continuous-time quantum Monte Carlo method in the hybridization-expansion limit (CT-HYB) was used to solve the auxiliary quantum impurity problem,[24] where the Ir t_{2g} orbital was chosen as our correlated subspace in a single-site DMFT approximation. For the CT-HYB calculations, up to 10^{10} Monte Carlo steps (at $T = 58$ K) were employed for each Monte Carlo run.

In most runs, the temperature was set to be 232 K, but in some calculations with trial antiferromagnetic orders T was lowered down to 58 K. We tried to stabilize three different types of collinear antiferromagnetic orders (Néel-, zigzag-, and stripy-types), but all tried magnetic orders did not remain stable and the paramagnetic order still sets in down to $T = 58$ K, which is rather unusual for the normal DMFT calculation as the DMFT result usually overestimates the ordering temperature, which may indicate an effect of the magnetic frustration of β-Li_2IrO_3.

The reasonable hybridization window of -10 to +10 eV (with respect to the Fermi level) was chosen, and $U = 5$ eV and $J_H = 0.8$ eV of on-site Coulomb interaction parameters were used for the Ir t_{2g} orbital. This values are slightly different compared to those used in another eDMFT study of layered perovskite and pyrochlore iridates,[25,26] $(U, J) = (4.5, 0.8)$ eV, but this difference is not expected to lead to a qualitative difference.

Values of U and J_H in eDMFT are significantly larger than those adopted in DFT+SOC+U calculations because of the different choice of projectors for the correlated subspaces in both methods. For the Coulomb interactions, a simplified Ising-type (density-density terms only) approximation was applied to reduce the Monte Carlo noise, and a nominal double counting scheme was used with $n_d = 5$ for the double counting correction.

Table II and III show the optimized crystal structures at ambient pressure (the orthorhombic structure) and high pressure (the monoclinic structure), starting from experimental structures, showing a reasonable agreement in both DFT+SOC+U and eDMFT results.

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